

Synthesis, Crystal Structure, and Magnetic Properties of $\text{Bi}_3\text{Mn}_4\text{O}_{12}(\text{NO}_3)$ Oxynitrate Comprising $S = 3/2$ Honeycomb Lattice

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Abstract: Neutron diffraction and FT-IR analysis revealed that the novel oxynitrate $\text{Bi}_3\text{Mn}_4\text{O}_{12}(\text{NO}_3)$ (space group $P3$, $a = 4.9692(1)$ Å, $c = 13.1627(3)$ Å) prepared by hydrothermal synthesis is of a new structural type including flat NO_3 layers alternating with blocks of two PbSb_2O_6 -like layers. Mn^{4+} ($S = 3/2$) forms a regular honeycomb lattice, and magnetic susceptibility data indicated two-dimensional magnetism. Despite its Weiss constant of -257 K, no long-range ordering was observed down to 0.4 K because of the magnetic frustration due to the competition between the nearest and the next-nearest antiferromagnetic interactions.

Introduction

The magnetic properties of a solid reflect the arrangement of the magnetic ions in its crystal structure. Low-dimensional antiferromagnets exhibit a variety of ground states depending on the spin number and the spin configuration,^{1,2} and the discovery of new $S = 1/2$ kagome antiferromagnet without lattice distortion³ and the confirmation of the absence of long-range ordering (LRO) in the $S = 1/2$ triangle antiferromagnet NiGa_2S_4 ⁴ have drawn interest to the frustrated two-dimensional magnets. Hydrothermal synthesis can produce crystal structures with cation environments and oxidation states unusual for high-temperature solid-state reaction conditions, and three new allotropic forms of bismuth oxide were prepared under hydrothermal conditions^{5–7} because this synthesis method stabilizes the distorted crystal structures due to the $6s^2$ lone pairs of Bi^{3+} and the covalent Bi–O bonds.

A further advantage of hydrothermal reaction is its ability to produce compounds otherwise unstable at high temperatures,

such as those containing water or other gas species. $\text{RBi}_2\text{O}_4\text{NO}_3$ ($R = \text{Y}, \text{Sm}, \text{Eu}, \text{Gd}, \text{Tb}, \text{Dy}, \text{Er}, \text{Yb}$) obtained by hydrothermal reaction comprises layers built of standing NO_3^- groups, where nitrogen is tetrahedrally coordinated by oxygen.⁸ Other bismuth oxynitrates, such as $\text{BiMO}_2(\text{NO}_3)$ ($M = \text{Sr}, \text{Ba}$), have flat NO_3^- groups parallel to the c axis but with bond angles different from 120° .⁹ Since NO_3^- groups separate the M–O networks in these compounds, bismuth oxynitrates with magnetic ions are expected to show two-dimensional magnetism. Planar and regular NO_3^- groups with 120° O–N–O angles are also expected to act as a template for forming magnetic ions into triangular and triangle-derived lattices where magnetic frustration takes place.²

In this paper we report the first example of bismuth–manganese oxynitrate $\text{Bi}_3\text{Mn}_4\text{O}_{12}(\text{NO}_3)$ prepared by the hydrothermal method, representing a new structural type comprising trigonal NO_3 flat layers alternating with PbSb_2O_6 -like layers. The bismuth(III) ions occupying three crystallographic sites have distorted octahedral and umbrella-like coordinations, and the edge-sharing MnO_6 octahedra form honeycomb lattices that are well separated by Bi and NO_3^- layers. Despite the relatively large Weiss constant of -257 K, no long-range antiferromagnetic (AF) ordering was found at temperatures down to 0.4 K because of the magnetic frustration due to the presence of next-nearest AF interaction.

Experimental Section

The sample powder was prepared from $\text{Mn}(\text{NO}_3)_2 \cdot 9\text{H}_2\text{O}$ and NaBiO_3 (Kanto Chemicals) by hydrothermal reaction—4 days at

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Table 1. Data Collection and Refinement Details for $\text{Bi}_3\text{Mn}_4\text{O}_{12}(\text{NO}_3)$

| radiation type | neutron |
|-----------------------------|---|
| Formula unit | $\text{Bi}_3\text{Mn}_4\text{O}_{15}\text{N}$ |
| Symmetry | Trigonal |
| Space group | $P3$ (No 143) |
| Z | 1 |
| Lattice parameters: | |
| a , Å | 4.9692(1) |
| c , Å | 13.1627(3) |
| Instrument | HRPD (JRR-3) |
| Wavelength | 1.82307 |
| Temperature | 300 K |
| Container material | vanadium |
| 2θ range | $12.5\text{--}145^\circ$ |
| 2θ step | 0.05° |
| Number of reflections | 272 |
| Number of variables | 42 |
| Profile function | Pseudo-Voigt |
| U | 0.29(6) |
| V | $-0.40(11)$ |
| W | 0.22(4) |
| Preferred orientation (hkl) | 100 |
| Secondary phases | MnO_2 6.7(1) w% |
| Agreement factors: | |
| R_p , % | 5.68 |
| R_{wp} , % | 7.3 |
| R_{Bragg} , % | 4.45 |
| R_f , % | 2.59 |
| χ^2 | 2.89 |
| G.o.f. | 2.9 |

533 K—in 70 mL Teflon-lined autoclaves. SEM pictures were taken using a JEOL JSM-6500F, and TEM and ED pictures were taken with a Topcon EM-002B. FT-IR measurements were made with a Jasco FT/IR-4100 apparatus using both ATR reflection mode and KBr transmission mode attachments. TG/DTA measurements accompanied by mass spectrometry results were obtained, in a helium atmosphere at a heating rate 5 K/min in the temperature range from 300 to 1173 K, by using a Rigaku ThermoPlus TG-8120 equipped with a Thermo Mass system. Synchrotron X-ray powder diffraction (SXR) measurements were made using beam-line BL02B2 at the SPring-8 facility. The data were collected with a constant wavelength ($\lambda = 0.422225$ Å) at room temperature. Neutron powder diffraction (NPD) measurements were made, using a high-resolution powder diffractometer (HRPD) at the JRR-3 reactor of the JAEA site in Tokai, at a constant wavelength ($\lambda = 1.82307$ Å) in a vanadium container at room temperature. EXPO2004¹⁰ and FOX programs¹¹ were used for crystal structure solution, and the FullProf&Winplot Suite¹² was used for refinement of the structure. The details of the data collection and of the structure solution and refinement are summarized in Table 1. Magnetic susceptibility was measured with a SQUID magnetometer (Quantum Design MPMS XL) in an external magnetic field of 100 Oe on cooling between 2 and 400 K. The specific heat of a pellet solidified at 6 GPa in a cubic-anvil-type high-pressure apparatus was measured at temperatures between 0.4 and 300 K by a relaxation method with a Quantum Design PPMS.

Results and Discussions

Synthesis and Crystal Structure. The hydrothermal reaction yielded a fine black powder consisting of particles no more than

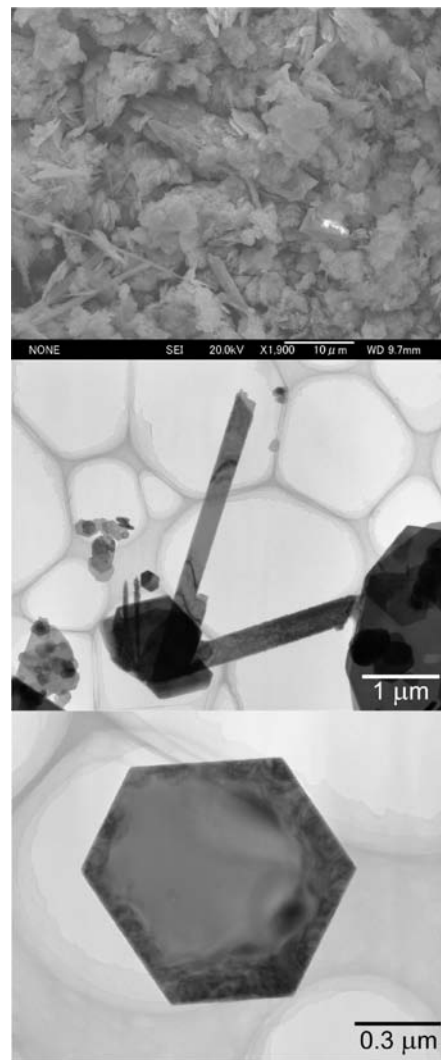


Figure 1. SEM (top) and TEM (middle) photos of the powder obtained by hydrothermal reaction. The bottom TEM image shows a particle of the main phase.

about one micrometer across (Figure 1). X-ray phase analysis revealed only a small amount of MnO_2 , and all of the other diffraction peaks seemed to be due to a new phase. TEM studies confirmed the presence of two morphologies: the major phase corresponding to a hexagonal platelet shape and the minor phase corresponding to a needle-like shape (Figure 1). Indexing of the electron diffraction patterns of the major phase provided the approximate dimensions of the hexagonal cell: $a \approx 5$ Å and $c \approx 13$ Å (Figure 2). As no systematic absences were observed, the space group had to be chosen from 16 possible groups. When analyzing the synchrotron X-ray data (not shown) with the lowest hexagonal symmetry group ($P6$), a combination of the direct method of structure solution from powder pattern implemented in EXPO2004 and the direct-space method implemented in FOX allowed us to locate heavy atoms in a structural model similar to PbSb_2O_6 (with bismuth in lead sites and with manganese in antimony sites) but having a c parameter approximately three times longer. The hexagonal cell includes two PbSb_2O_6 -like layers separated by a light-atoms layer unsolved from X-ray data. Since the presence of NO_3^- groups in the structure was suggested by FT-IR measurements that showed clear signal with the wavenumber 1380 cm^{-1} (Figure 3), similar to the signal seen in FT-IR spectroscopy results for

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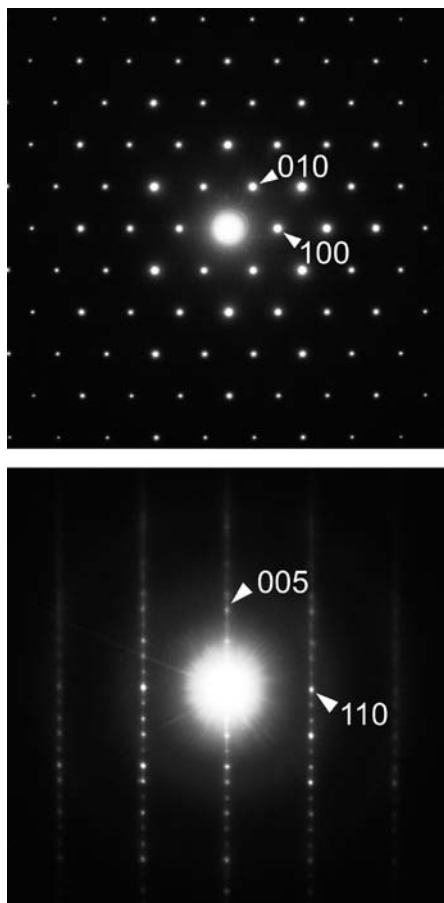


Figure 2. Electronic diffraction patterns of hexagonal Bi₃Mn₄O₁₂(NO₃).

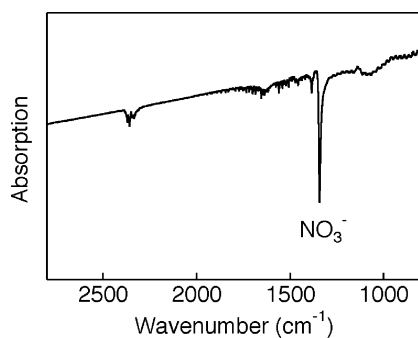


Figure 3. Bi₃Mn₄O₁₂(NO₃) FT-IR spectra measured in the transmission mode.

YBi₂O₄NO₃,⁸ the neutron diffraction data were analyzed on a model with NO₃⁻ groups between the two PbSb₂O₆-like layers. TG/DTA mass spectroscopy confirmed the presence of nitrogen ions as discussed later. Figure 4 shows the NPD pattern with the result of the Rietveld refinement. Further reduction of the space group to *P*3 was necessary to place the flat NO₃⁻ group whose presence was indicated by the direct-space analysis of the neutron diffraction data. The absence of inversion symmetry was verified by means of optical second harmonic generation measurement.

The refined crystal structure is illustrated in Figure 5. The structural parameters refined from neutron data are listed in Table 2. Since no sign of off-stoichiometry was observed, the occupancy factors were fixed to unity. The refinement showed the presence of 6.7 wt % MnO₂ as a secondary phase. To our

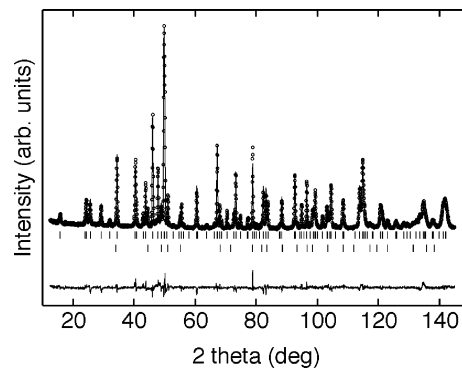


Figure 4. Rietveld plot of neutron data for Bi₃Mn₄O₁₂(NO₃). Observed (circles), calculated (line) and difference (lower line) profiles are shown together with Bragg markers for the Bi₃Mn₄O₁₂(NO₃) phase and MnO₂.

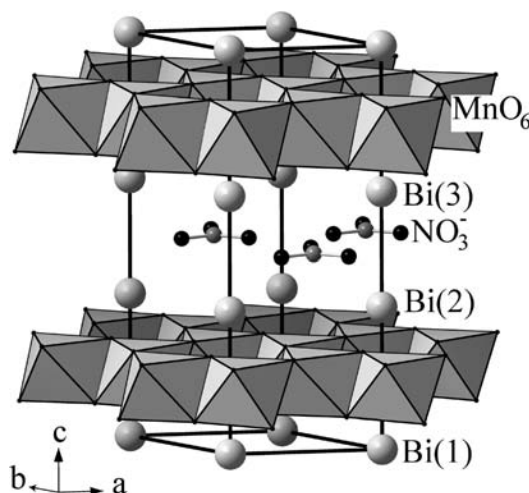


Figure 5. Polyhedral representation of Bi₃Mn₄O₁₂(NO₃) structure.

Table 2. Positional and Thermal Parameters for Bi₃Mn₄O₁₂(NO₃) as Obtained from Neutron Data

| atom | Wyck. | fraction | x | y | z | B ₈₀ (Å ²) |
|-------|-------|----------|----------|----------|----------|-----------------------------------|
| Bi(1) | 1a | 1.0 | 0 | 0 | 0.025(4) | 0.61(5) |
| Bi(2) | 1a | 1.0 | 0 | 0 | 0.673(5) | 0.61 |
| Bi(3) | 1a | 1.0 | 0 | 0 | 0.382(5) | 0.61 |
| Mn(1) | 1c | 1.0 | 2/3 | 1/3 | 0.855(5) | 0.35(11) |
| Mn(2) | 1b | 1.0 | 1/3 | 2/3 | 0.852(6) | 0.35 |
| Mn(3) | 1c | 1.0 | 2/3 | 1/3 | 0.218(5) | 0.35 |
| Mn(4) | 1b | 1.0 | 1/3 | 2/3 | 0.223(6) | 0.35 |
| O(1) | 3d | 1.0 | 0 | 0.381(4) | 0.927(5) | 0.75(5) |
| O(2) | 3d | 1.0 | 0 | 0.623(1) | 0.762(5) | 0.75 |
| O(3) | 3d | 1.0 | 0 | 0.360(4) | 0.295(5) | 0.75 |
| O(4) | 3d | 1.0 | 0 | 0.626(3) | 0.142(5) | 0.75 |
| O(5) | 3d | 1.0 | 0.475(6) | 0.522(6) | 0.537(4) | 2.56(14) |
| N | 1b | 1.0 | 2/3 | 1/3 | 0.535(5) | 0.51(16) |

knowledge, the novel bismuth manganese oxynitrate represents a new structural type. Selected interatomic distances are listed in Table 3, and the bond valence sums summarized in Table 4 indicate the following valence state: Bi³⁺₃Mn⁴⁺₄O²⁻₁₂(NO₃)⁻. The Mn⁴⁺ oxidation state is consistent with the formation of MnO₂ as the secondary phase. The manganese atoms have octahedral environment, slightly compressed along the *c* axis of the unit cell. The edge-sharing MnO₆ octahedra form honeycomb lattices as schematically illustrated in the inset of Figure 7b. The bismuth atoms, on the other hand, are in distorted environments. Bi1 is also in an octahedron but is shifted from the center. Bi2 and Bi3 have umbrella-like coordinations. They

Table 3. Selected Interatomic Distances for $\text{Bi}_3\text{Mn}_4\text{O}_{12}(\text{NO}_3)$

| bond | r (Å) | Bond | r (Å) |
|---------------------------|---------|---------------------------|---------|
| Bi(1)–O(1) ($\times 3$) | 2.29(5) | Mn(1)–O(1) ($\times 3$) | 1.82(5) |
| Bi(1)–O(4) ($\times 3$) | 2.41(6) | Mn(1)–O(2) ($\times 3$) | 2.00(6) |
| Bi(2)–O(2) ($\times 3$) | 2.18(5) | Mn(2)–O(1) ($\times 3$) | 1.84(6) |
| Bi(2)–O(5) ($\times 3$) | 3.06(5) | Mn(2)–O(2) ($\times 3$) | 1.97(6) |
| Bi(2)–O(5) ($\times 3$) | 3.08(5) | Mn(3)–O(3) ($\times 3$) | 1.89(5) |
| Bi(3)–O(3) ($\times 3$) | 2.12(5) | Mn(3)–O(4) ($\times 3$) | 1.86(5) |
| Bi(3)–O(5) ($\times 3$) | 3.22(6) | Mn(4)–O(3) ($\times 3$) | 1.86(5) |
| Bi(3)–O(5) ($\times 3$) | 3.23(6) | Mn(4)–O(4) ($\times 3$) | 1.89(6) |
| N–O(5) ($\times 3$) | 1.23(4) | | |

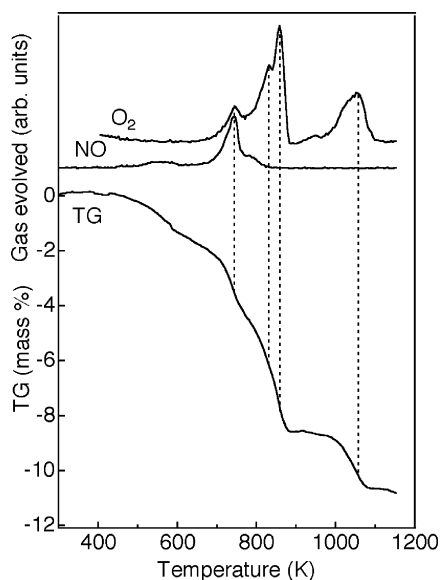
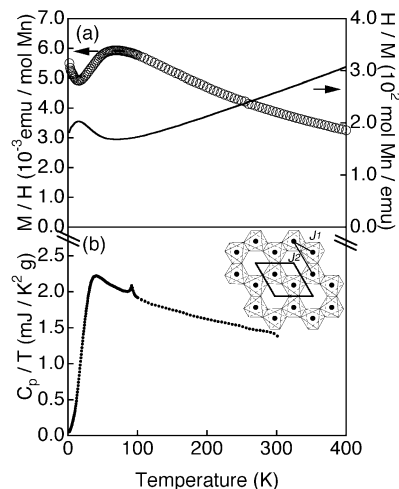
Table 4. Bond Valence Sums for $\text{Bi}_3\text{Mn}_4(\text{NO}_3)\text{O}_{12}$

| Bi1 | Bi2 | Bi3 | Mn1 | Mn2 | Mn3 | Mn4 | O1 | O2 | O3 | O4 |
|-----|-----|-----|-----|-----|-----|-----|-----|-----|-----|-----|
| 3.0 | 2.8 | 3.1 | 4.1 | 4.0 | 4.3 | 4.3 | 2.2 | 1.8 | 2.4 | 1.9 |

have three short Bi2(3)–O2(3) bonds on one side and six long Bi2(3)–O5(5) bonds on the other side. These environmental distortions might be explained by the space needed for the lone pairs of Bi^{3+} . The distance between PbSb_2O_6 -like layers and NO_3 layers is quite large: the Bi2–O5 distance is 3.1 Å and the Bi3–O5 distance is 3.2 Å. Since each of these distances is significantly greater than the sum of the ionic radii, the bonding should be rather weak.

Thermal Analysis. The existence of nitrogen atoms in the structure was confirmed by the TG measurements accompanied by mass spectroscopy. When the structure is heated in helium it loses NO and O_2 at 723 K and loses more O_2 at 830, 870, and 1050 K (Figure 6). This suggests that $\text{Bi}_3\text{Mn}_4\text{O}_{12}(\text{NO}_3)$, like $\text{YBi}_2\text{O}_4\text{NO}_3$,⁸ releases the NO_3 group at 723 K, but this suggestion needs to be confirmed by *in situ* XRD study.

Magnetic Properties. Because the $\text{Bi}_3\text{Mn}_4\text{O}_{12}(\text{NO}_3)$ pellet solidified at 6 GPa was highly resistive, the system can be regarded as a localized spin system. The temperature dependence of the magnetic susceptibility shown in Figure 7a shows a broad maximum centered at about 70 K. This broad maximum is a characteristic feature of a low-dimensional antiferromagnet and is consistent with the two-dimensional crystal structure. Fitting the data between 300 and 400 K to the Curie–Weiss law $\chi = C/(T-\theta) + \chi_0$ where C is the Curie constant, θ is the Weiss

**Figure 6.** TG and mass spectroscopy data for $\text{Bi}_3\text{Mn}_4\text{O}_{12}(\text{NO}_3)$ in helium.**Figure 7.** Temperature dependence of the magnetic susceptibility (open circles) and the inverse susceptibility (solid line) (a) of $\text{Bi}_3\text{Mn}_4\text{O}_{12}(\text{NO}_3)$. Total specific heat divided by temperature (b) of $\text{Bi}_3\text{Mn}_4\text{O}_{12}(\text{NO}_3)$ solidified at 6 GPa. Inset shows the network of Mn spins.

constant, and χ_0 is the temperature-independent term gave $C = 2.21$, $\theta = -257$ K, and $\chi_0 = -1.16 \times 10^{-4}$ emu/mol. This Curie constant is as expected for a Mn^{4+} ($S = 3/2$) system with $g = 2.17$ and is thus consistent with the $4+$ oxidation state determined for Mn in the structure analysis. Despite the large AF Weiss constant of -257 K, no sign of LRO was observed. The increased susceptibility seen below 20 K is not an indication of the LRO because no corresponding anomaly is evident in the specific heat data (Figure 7b), and it should be attributed to the presence of an undetected impurity phase or to free spin due to the structural defects. Figure 7b shows the temperature dependence of the specific heat divided by the temperature (C_p/T). Since there is no nonmagnetic reference, the lattice contribution is not subtracted. The small peak at 95 K corresponds to the AF ordering of MnO_2 detected by the SXRD and PND measurements. No other peak is found between 0.4 and 300 K, confirming the absence of the LRO in $\text{Bi}_3\text{Mn}_4\text{O}_{12}(\text{NO}_3)$. The C_p/T data show a broad maximum at around 40 K and decrease to zero with decreasing temperature. This suggests the presence of a short-range ordering.

$\text{Bi}_3\text{Mn}_4\text{O}_{12}(\text{NO}_3)$ showed a two-dimensional magnetism consistent with the determined crystal structure. Because of its 3-fold symmetry, the honeycomb lattice of Mn^{4+} ($S = 3/2$) is uniform, without any distortion. It should also be pointed out that the orbital ordering as proposed for a $S = 1/2$ Kagome compound $\text{Cu}_3\text{V}_2\text{O}_7(\text{OH})_2 \cdot 2\text{H}_2\text{O}$ ¹³ does not take place in this compound because of the d^3 nature of Mn^{4+} ion. Despite the AF Weiss temperature of -257 K, no LRO was observed down to 0.4 K. The ground state of an ideal AF honeycomb lattice system with only the AF nearest-neighbor interaction is LRO because the system is free from frustration. If next-nearest interaction is also AF, however, the system is frustrated.¹⁴ In the present compound the nearest AF interaction mediated by the orthogonal Mn–O–Mn bond is rather weak. Considering the Weiss temperature of -257 K and the number of the nearest-neighbor spins, 3, one would expect the exchange interaction to be smaller than 100 K. The next-nearest interaction mediated by the Mn–O–O–Mn bond is therefore not negligible. The

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current compound is the first example of frustrated honeycomb antiferromagnet without long-range ordering. The ground states of such systems with classical¹⁴ and quantum¹⁵ spins had been investigated theoretically, but no actual compound had been known. It was predicted that the $S = 3/2$ honeycomb system with $J_2/J_1 > 0.16$ would not exhibit a long-range ordering.¹⁴ Further quantitative analyses of the susceptibility and specific heat data are desirable. It would also be interesting to prepare Bi₃M₄O₁₂(NO₃) with other M cations such as Cr⁴⁺ ($S = 1$) or V⁴⁺ ($S = 1/2$) in order to investigate the magnetism of frustrated honeycomb lattices with other spin numbers.

Conclusions

Hydrothermal conditions with nitrate ions stabilize a new type of structure: Bi₃Mn₄O₁₂(NO₃) (space group $P3$, $a = 4.9692(1)$ Å, $c = 13.1627(3)$ Å) containing flat and regular layers of NO₃ composition alternating with blocks of two PbSb₂O₆-like layers.

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The present system is a good model compound of an $S = 3/2$ honeycomb antiferromagnet. Despite the relatively large Weiss constant of -257 K, no AF LRO was found down to 0.4 K because of the magnetic frustration due to the presence of next-nearest AF interaction.

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